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RESEARCH ON ADDITIVE MANUFACTURING (AM) TECHNOLOGY DEVELOPMENT FOR RADIO FREQUENCY QUADRUPOLE (RFQ) FABRICATION AND PERFORMANCE IMPROVEMENT

**PĒTĪJUMS PAR ADITĪVĀS RAŽOŠANAS (AM) TEHNOLOGIJAS
IZSTRĀDĀŠANU RADIOFREKVENCES KVADRUPOLA (RFQ)
IZGATAVOŠANAI UN VEIKTSPĒJAS UZLABOŠANAI**

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Additive manufacturing is a relatively new technology with significant advances over conventional (subtractive) machining methods. One of the most known is minimum material waste and unbeatable potential to build conventionally inaccessible structures like complex internal channels and lattice structures. Therefore, a compact linear particle accelerator (RFQ) became an excellent candidate for demonstrating AM potential to the accelerator community. It is essential to mention that RFQs conventionally are manufactured from high-purity electronic oxygen free copper (OFE-Cu) bars, but in the case of AM, primarily electronic tough pitch copper (ETP-Cu) [1]. Indeed, according to S. Gruber, ETP-Cu for additive manufacturing gives similar or even better results than OFHC-Cu.

In the last decade, developments in the metal powder bed fusion induced by the “green” laser beam application made a significant step toward highly reflective material processing. The TRUMPF SE+Co KG “green” laser system presented in November 2018 [2] proved its reliability and efficiency for pure copper (ETP-Cu) processing. Therefore, the question about AM application for accelerator components became logical. Moreover, “green” laser systems have already been tested, which allowed us to be confident that conventional manufacturing technology will soon be replaced. However, already at the

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beginning of I.FAST WP 10.2 it was clear enough that postprocessing methods would be needed to achieve good “skin layer” quality for the 750 MHz system. To ensure low-value power loss in the “skin layer”, the average surface profile roughness value was initially targeted to $R_a = 0.2 \mu\text{m}$.

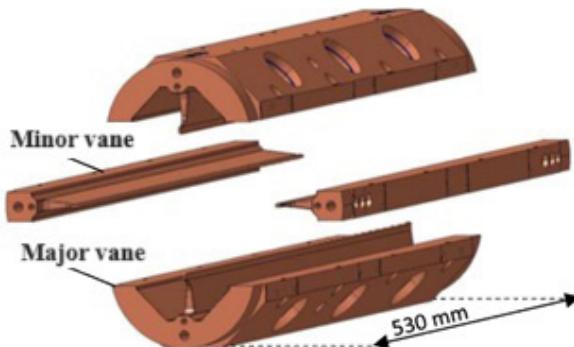


Fig. 1. Exploded view of PIXE-RFQ module assembly, two major vanes and two minor vanes [3].

Conventionally, a section of a radio frequency quadrupole is manufactured from four separate pieces using a relatively long list of different manufacturing operations, more than 50. CERN's EN-MME and ATS teams developed the compact RFQ design for conventional manufacturing technology and manufacturing procedure, and S. Mathot guided manufacturing. An exploded view of one RFQ section is shown in Fig. 1 to give an impression of a conventional design. 5-axis milling operations are dominant during CNC machining. Conventional technology requires high material volume removal, and after each milling operation, the workpiece must be placed in a vacuum furnace for internal stress relief.

Furthermore, to be able to control the manufacturing process after each milling and stress relief, CMM measurements are recommended. These highly trackable solutions are necessary to provide vane tip modulation tolerances in the range of $5 \mu\text{m}$ for each machined part in the vane tip modulation area. In total, manufacturing time for the PIXE accelerator took approximately 16 months.

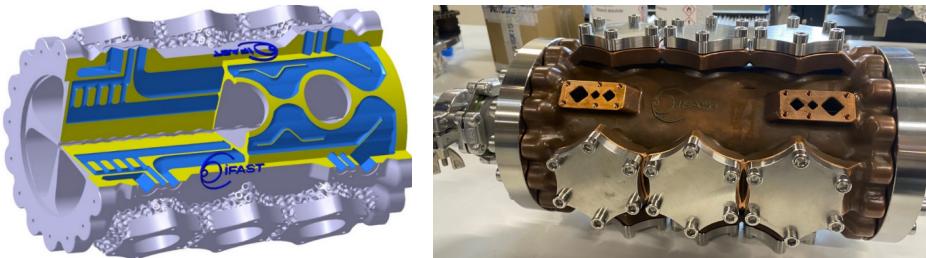


Fig. 2. Single body AM built 750 MHz RFQ prototype model and post-processed prototype prepared for vacuum tests in IJCLab, Orsay.

To summarize the initial research on AM technology for compact-size RFQ, several points must be emphasized:

AM is a highly progressing technology well suited for the complex geometries of accelerators like RFQs. AM technology has a high potential to replace conventional manufacturing methods in the near future.

AM allows improvement of the shape and routing of the RFQ cooling channels, improving heat removal and thermal distortions significantly. With a 1.5 mm wall thickness allowance, an average 32 % improvement was reached.

Surface postprocessing for AM parts is critical, and compensation methods must be applied in the design or manufacturing phase. PBF-LB/M/ETP-Cu, “as build” part average surface roughness profile values, are $\sim Ra = 15 \mu\text{m}$ and $Rz = 100 \mu\text{m}$, which is far from $Ra = 0.2 \mu\text{m}$.

Acknowledgements

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EXPERIMENTAL STUDIES ON THE PERFORMANCE OF CARBIDE TOOL IN STAINLESS STEEL TURNING PROCESS

CIETSĀKAUSĒJUMA GRIEZĒJINSTRUMENTA VEIKTSPĒJU
EKSPERIMENĀLIE PĒTĪJUMI NERŪSĒJOŠĀ TĒRAUDA
VIRPOŠANAS PROCESĀ

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The aim of the represented study was to investigate the accuracy of the stainless steel turning process depending on the wear of Al_2O_3 nano-coated cutting tools. Modern, high wear-resistant coating provides longer tool life in the machining process (Fig. 1). This will make it possible to find out the effect of increased technological processing parameters (cutting speed, feed and cutting tool setting angle) on the turning result of the machinability of two stainless steels (AISI 420 and AISI 304) – martensitic and ferritic, with the surface roughness parameter Ra when using new cutting tools covered using Duratomic-technology (Fig. 2), compare it to these applied tools and develop a mathematical model for determining the roughness (Ra) of the processed surface.

To achieve the goal of the study, the following tasks have been determined:

- to perform a comprehensive analysis of selected processing technologies and applied tools;
- to perform turning process experiments and analysis of the results, clarifying the dependence of the roughness (Ra) of the turned surface on increased processing modes: cutting speed, feed and cutting edge angle;
- to compare the results of the turned surface of two stainless steel grades – AISI 420 and AISI 304;
- to compare the influence of tools produced by different manufacturers when processing blanks of two stainless steel grades (AISI 420 and AISI 304) at increased cutting speeds;

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- to perform an analysis of the tool wear mechanism using increased processing modes;
- to perform an analysis of the chip formation process using increased cutting speeds [1].

EXPERIMENTAL PROCESS

A lathe, an AISI 304 stainless steel workpiece with an austenitic structure, and an AISI 420 stainless steel workpiece with a martensitic structure were used to realize the experimental part.

The new cutting tools of *Seco*, *Walter*, *Sumitomo*, and *Kennametal* were used, which are characterized by a wide application area for various stainless steel brands and structures.

Data of the Ra value results of the processed surface roughness profile obtained in the practical experiment were processed using a computer, implementing a 3-factor experimental plan.

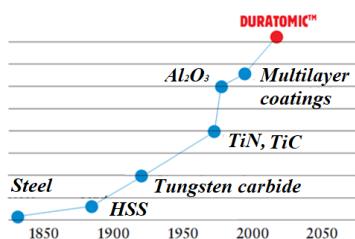


Fig. 1. Progress of cutting tool coatings [2].

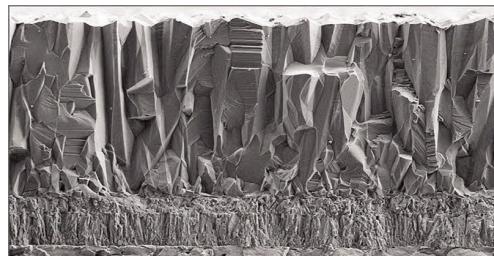


Fig. 2. Cutting tool coated with Duratomic technology, the vertical crystal positioning structure of the Al_2O_3 coating is visible [2].

When examining all the results of the conducted experiments, it can be seen that the tools of different manufacturers are not intended for use at increased cutting modes.

When applying cutting tool nano-coatings, it is important to compare the wear resistance of cutting tools. The following cutting tools with coatings were chosen for the wear comparison experiment: KC730, TM4000, AC200, CP500, WPP200, and TP3500.

The tool wear was measured on the cutting surface and flank of the insert tip and after the cutting process for each combination of turning parameters.

As a result of the experiments, it was found that depending on the cutting conditions, different tool geometries and types of coatings, and the characteristics of the processed

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material (grade 304 and 420 stainless steel), the dominant wear was observed on the major flank, hm, and on the chip surface shown in cutting tool geometry figures in [1].

KC730 (Kennametal) showed a higher wear result. This tool, due to the rather uncomplicated chip breaker geometry, demonstrated better turned surface results than the other tools, but the wear values were very high. It is interesting that this tool, when turning two different grades of stainless-steel groups, showed different wear – minimum wear when turning AISI 420 and maximum wear when turning AISI 304 stainless steel. The data was obtained by reading the wear values after a complete machining cycle not exceeding the expected service life of the tool (15 min).

PRACTICAL APPLICATION

The results of the given research are necessary for the needs of Seco Tools AB instrument manufacturers and the local representatives of this company (the acceptance review was received from the SECO Baltic office).

Cutting tools coated with Duratomic technology have been proven to last longer than described in catalogues, creating great potential for increased productivity.

The results of the study have been presented in several scientific articles written during the development of the PhD Thesis and have been cited in various publications of several scientists and in Master and Doctoral Theses.

Conclusions

1. From the obtained data, it can be concluded that TM4000 cutting tools coated with Duratomic technology are more wear-resistant. During the processing, vibrations occur, which deteriorate the quality of the turned surface.
2. Using higher cutting speeds, without cooling emulsions, the highest service life variant was achieved with Duratomic TM4000 coating (Seco), which showed minimal cutting-edge wear at 50 % increased cutting speeds, providing good (judging from previous calculations and compared to other manufacturer's tools) machined surface quality (in such a condition the tool could still be used).
3. Comparing the selected coatings, we can admit that from the point of view of wear, the coating made with Duratomic technology is considered the best because it causes less wear to the tool.

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DECONVOLUTION MODELS FOR EVALUATING THE TRUE MICROHARDNESS OF THIN NANOSTRUCTURED TRIBOLOGICAL COATINGS DEPOSITED BY PVD TECHNIQUE

AR PVD METODI IEGŪTO PLĀNO NANOSTRUKSTURĒTO
TRIBOLOGISKO PĀRKLĀJUMU PATIESĀS MIKROCIETĪBAS
NOVĒRTĒŠANA, IZMANTOJOT MATEMĀTISKOS
DEKONVOLŪCIJAS MODEĻUS

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This study discusses micromechanical properties and true microhardness determination of nanostructured tribological coatings (NTC) based on a multilayered alternating nitride/carbonitride bilayer substructure of transition metals. The constituent nitride/carbonitride bilayers in the superlattice structure of NTC were alloyed with refractory metals, denoted as Me = Me1 or Me2 = Cr, Hf, Nb and Zr. The resulting NTC coatings were deposited onto 100Cr6 steel substrates using the advanced physical vapour deposition (PVD) technique denoted here as the high-power impulse plasma magnetron sputtering (HiPIPMS).

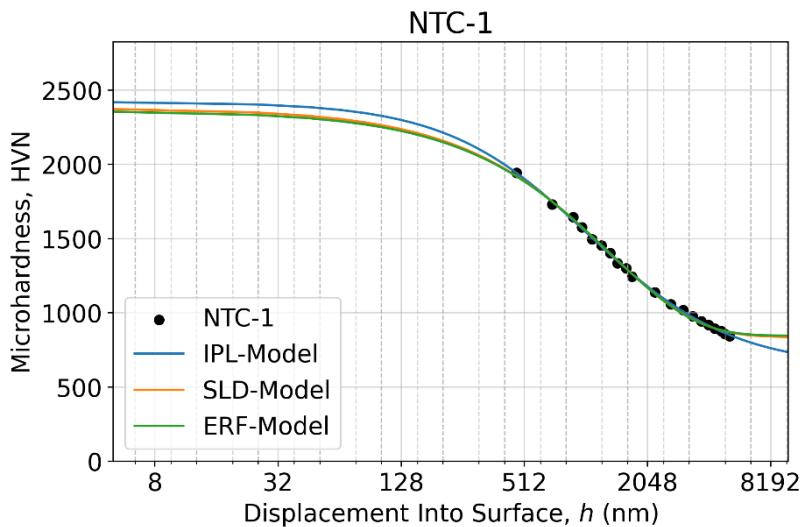


Fig. 1. Microindentation response as apparent microhardness – displacement, $H-h$, curve of the NTC-1 sample calculated using measured two diagonals of the Vickers imprints at each specified test load within (10–700) gf-range (0.098–6.86 N) in semi-log scale. The fitted microhardness-displacement curves relevant to the proposed three sigmoidal decay models are shown as solid lines.

Comprised crystalline nanometer-scale TiAlSiMe1-N/TiMe2-CN nanoparticles strengthened by Me additives significantly increased the NTC micro-hardness to over 3200 HV. The primary focus of this research was to determine the true microhardness of the NTC film samples. The apparent microhardness (Ha) of the film/substrate system for various NTC samples was measured during microindentation testing using the Vickers method. Nine NTC samples were tested, each generating a corresponding microindentation dataset containing between 430 and 640 imprints, depending on the specific NTC sample. These datasets were analyzed using three distinct empirical approaches: (i) the inverse power-law model (IPL-Model), (ii) the sigmoid-like decay model (SLD-Model), and (iii) the error function model (ERF-Model). The observed solid correlation between the proposed models and experiments suggests that the true microhardness estimates (H_f) obtained through the empirical mathematical modeling approach are reliable.

Acknowledgements

This research was funded by the Latvian Council of Science, project “Carbon-rich self-healing multifunctional nanostructured smart coatings (NSC) for high-tech applications using high-power confined plasma technology for their deposition”, No. 2019/1-0385.

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DETERMINATION OF SERVICE LIFE OF COATED FRICTION PAIR PARTS

BERZES PĀRA DETAĻAS AR PĀRKLĀJUMU
KALPOŠANAS LAIKA NOTEIKŠANA

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The structural safety of the machine depends on the material chosen for the machine parts, the dimensions of their shape and the surface finish – roughness. In all cases, the structural safety must be justified by appropriate calculations. In many cases, machine parts do not break but wear down through the friction process, changing dimensions, and need to be replaced – repaired. To characterize the wear process, we can use the so-called linear wear rate according to the wear fatigue theory.

$$I_h = \frac{h}{L_b} = \frac{h}{V \cdot t'}$$

where

h – linear wear of material;

L_b – wear path length;

V – linear velocity;

t' – life time of wear parts.

The linear intensity of wear, I_h , can be calculated theoretically using analytical relationships. The wear rate depends on the load, the physical-mechanical and frictional parameters of the material and the surface roughness – surface texture – of the friction parts. The abrasion rate varies from 10^{-6} to 10^{-12} for different parts. The analytical calculation is time-consuming and sufficiently accurate if all the parameters entered

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in the formulae can be determined accurately. Experimental determination of the wear rate is obviously more accurate, but it is a labour-intensive process.

$$t = \frac{h_{max}}{V \cdot I_h},$$

where

h_{max} – permissible maximal linear wear of material;

I_h – wear path length.

By knowing the wear rate and substituting h_{max} for h , the maximum allowable wear, we can determine how long it will take to replace worn friction parts, thus predicting the lifetime of the equipment until repairs are required.

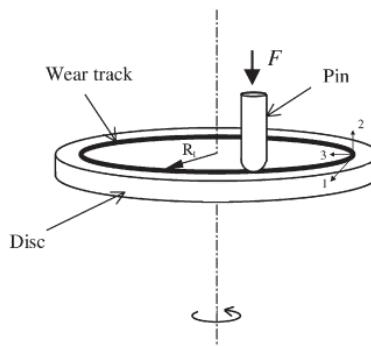


Fig. 1. Schematic illustration of pin-on-disc wear test.

Theoretically, I_h can be calculated using mathematical relationships given in [1]–[3].

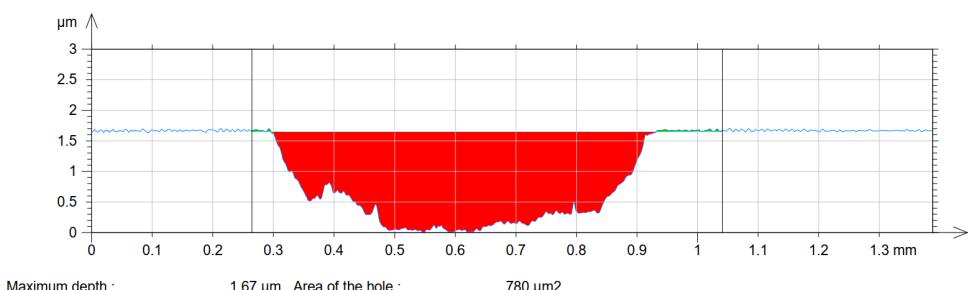


Fig. 2. Example of wear track geometry after tribological test.

The parameter h_{max} can be obtained experimentally by performing a tribotest with a pin-on-disc type tribometer Fig. 1, after the wear test by measuring with a profilometer and obtaining h_{max} at certain test conditions. The tribotrack profilogram is shown in Fig. 2.

Acknowledgements

This research was funded by the Latvian Council of Science, project “Carbon-rich self-healing multifunctional nanostructured smart coatings (NSC) for high-tech applications using high-power confined plasma technology for their deposition”, No. 2019/1-0385.

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STUDIES OF EXPERIMENTAL RESULTS AND ANALYTICAL CALCULATIONS OF WEAR OF FRICTION PAIR "STEEL-ANTI- FRICTION MATERIAL"

BERZES PĀRA "TĒRAUDS-ANTIFRIKCIJAS MATERIĀLS" NODILUMA
EKSPERIMENTĀLO REZULTĀTU UN ANALĪTISKO APRĒKINU
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Nowadays, the production of parts for mechanical engineering and other industries uses both the latest technologies and new material combinations with the aim to achieve longer product life. In order to increase the service life of mechanisms which use friction pairs and to renew surfaces worn out by friction pairs, not only traditional and long-established but also modern technologies are currently being used, such as surface treatment and hardening using laser and electron beam treatment, various types of surface heating methods, ion-plasma vacuum treatment, etc.

In addition to solutions for improving the performance of the surfaces, it is also necessary to be able to predict the service life of the parts in order to optimize the production process and make the necessary adjustments to the manufacturing process. Although several techniques and calculation methodologies are currently known that allow wear to be determined analytically [1]–[5], the approach of predicting wear based on lengthy experiments, which is both time-consuming and costly, still remains popular [6]–[8]. This is due to the fact that the analytical wear calculation would need to include the majority of parameters that exist in the actual wear process, which in most cases, is not done. This can be concluded when analyzing the analytical calculations available in the literature sources.

The analytical prediction of the service life of friction pairs, which is valid for practical engineering tasks and includes parameters that can be determined by modern measurement methods without the need for lengthy and resource-intensive experiments, not only speeds up the product design process but also makes a significant contribution to sustainable development.

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This study offers a wear calculation equation for linear wear of sliding friction pair steel-anti-frictional material (in this case, it is bronze), including standardized 3D roughness parameters, modelling surface microtopography with a separate section of probability theory – random field theory, and using fatigue theory for friction surface destructions:

$$U_n = 32 \cdot \frac{m!}{N_0 \cdot k_q^2} \cdot \frac{E^{m-2}}{\sigma_0^m} \cdot \frac{Sa^{m-1}}{RSm_1^{m-2}} \cdot \left(\frac{\pi^2}{\sqrt{2} \cdot K(e)^{1/2}} \right)^m \cdot q^2 \cdot \frac{L_b}{RSm_2^a},$$

where

m – degree of material fatigue curve equation;

N_0 – number of durability cycles of the material;

k_q^2 – coefficient depending on surface anisotropy;

E – modulus of elasticity of the material, MPa;

σ_0 – limit of resistance of the material under an asymmetric load cycle MPa;

Sa – arithmetical mean height, mkm;

RSm_1 – mean peak width for wearing component, mkm;

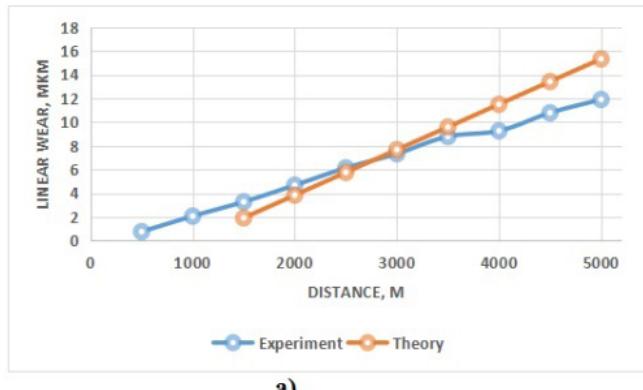
$K(e)^{1/2}$ – elliptic integral;

q – distributed load, MPa;

L_b – distance, m;

RSm_2^a – mean peak width for the component which causes wear.

To compare the analytical calculations of wear with the experimental results, the experiment was carried out on the pin-on-disc tribometer. The tested materials – pin (steel, 100Cr6 ISO) and disc (bronze, CuSn8 DIN). The theoretical and experimental wear graphs are shown in Fig. 1 a). The parameters applied for the experiment and calculations are shown in Fig. 1 b).



Parameter	Value	Unit
q	0.87	MPa
L_b	4000	m
m	4	-
σ_0	89	MPa
N_0	5×10^6	-
E	110000	MPa
S_a	0.6	μm
RS_{m1}	60	μm
RS_{m2}^α	70	μm
k_q	0.23	-
$K(e)$	500	-

b)

Fig. 1. a) Theoretical and experimental wear lines;
b) parameters necessary for calculations and experiment.

As can be seen from the graph, the experimental studies showed a close agreement with the results of the analytical calculations. Thus, the offered wear calculation equation could be used for practical engineering calculations without time consuming and rather expensive experiments for this combination of materials. To increase the reliability of results and make necessary improvements, future studies need to focus on comparing experimental and analytical wear values for other materials.

Acknowledgements

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SPIEDTVERTNES UN SISTĒMAS ŪDENĀRAŽA UZGLABŠANAI. KOMPONENTI UN RISINĀJUMI

PRESSURE TANK AND SYSTEMS FOR HYDROGEN STORAGE.
COMPONENTS AND SOLUTIONS

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Ūdeņraža degvielas sistēmām galvenais izaicinājums ir neliels ūdeņraža īpatnējais svars.

Barošanas sistēmām ir trīs galvenie elementi: spiediena tvertnes; kompresori; regulējošais aprīkojums.

Lielais spiediens (350 bar – smagajam kravas transportam, 700 bar – vieglajam transportam) nosaka prasības tvertnēm. Ir iespējami vairāki risinājumi:

I tips. Tērauda baloni, kuros tiek uzglabāta ūdeņraža gāze augstā spiedienā (līdz 200–300 bar). Lielākais trūkums ir balona lielais īpatsvars, piemēram, ja 50 l balons sver 60 kg, tad pie spiediena 200 bar tajā ir tikai 750 gramti ūdeņraža.

II tips. Baloni ar plānu metāla iekšējo apvalku ar daļēju ārējo kompozīta pārklājumu. Tie ir vieglāki nekā I tipa baloni (spiediens var sasniegt pat 500 bar), tos var izmantot ūdeņraža degvielas šūnu barošanai transportā.

III tips. Atšķirība no II tipa plānu metāla iekšējo apvalku pilnība pārklāj kompozīta slānis, kas tos padara vēl vieglākus nekā II tipa cilindrus.

IV tips. Baloni ar termoplasta iekšējo apvalku, kas no ārpusē tiks nostiprināti ar grafita šķiedru tinumu, tas ļauj palielināt spiedienu līdz 800–900 bar.

V tips. Baloni, kas izgatavoti tikai no viena grafita šķiedra slāņa.

Turklāt IV un V tipa baloni var izturēt iekšējo vakuumu un ir pilnībā pakļauti utilizācijai.

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Ir zināms vēl zemspiediena risinājums metāla hibrīdu uzglabāšanai, kas ļauj absorbēt un atbrīvot ūdeņraža gāzi. Zemspiediena risinājumu lieto stacionārajām un nelielām pārnēsājamām energijas sistēmām.

Ūdeņradi var pārveidot par šķidrumu ļoti zemā temperatūrā (aptuveni -253 °C) un uzglabāt kriogēnās tvertnēs. Šķidra ūdeņraža uzglabāšana galvenokārt tiek lietota lielam industriālam vai kosmiskam pielietojumam.

Lai veiksmīgi strādātu ar ūdeņraža degvielas sistēmām, ir svarīgi iekļaut visus nepieciešamos komponentus, kā arī ievērot drošības un ekspluatācijas noteikumus:

- 1) augsta spiediena ūdeņraža baloni aprīkoti ar speciāliem noslēgvārstiem, kas ļauj kontrolēti atbrīvot ūdeņraža gāzi;
- 2) spiediena regulatori tiek izmantoti, lai samazinātu augsto spiedienu no balona līdz drošam līmenim degvielas šūnā. Tie palīdz uzturēt vajadzīgo ūdeņraža gāzes plūsmu pie nemainīga spiediena šūnā;
- 3) tvertnēs ir uzstādītas drošības ierīces, piemēram, drošības vārsti, lai nepielautu pārmērīgu spiedienu; tas palīdz novērst balona bojājumus un eksploziju;
- 4) lai nepārtraukti uzraudzītu ūdeņraža koncentrāciju apkārtējā vidē, tiek izmantoti gāzes detektori;
- 5) tvertnes uzglabāšanas statji un nostiprinājumi. Augsta spiediena ūdeņraža baloni bieži tiek apvienoti, lai palielinātu ūdeņraža daudzumu, tāpēc balonus nostiprina īpašos statņos, lai novērstu nokrišanu vai bojājumus. Pareiza uzglabāšana novērš fiziskus balonu bojājumus un minimizē noplūžu risku;
- 6) vietās, kur tiek uzglabātas augsta spiediena ūdeņraža tvertnes, jāuzstāda ugunsdzēšanas sistēmas, lai ātri reaģētu potenciālās ugunsgrēku situācijās.

Augsta spiediena ūdeņraža tvertnes prasības var atšķirties atkarībā no reģiona un nozares. Lai novērstu nelaimes gadījumus un nodrošinātu drošu ūdeņraža gāzes izmantošanu, ir svarīgi ievērot drošības vadlīnijas un noteikumus. Eiropā tiek izmantoti cikliskie testi līdz 95 MPa un hidrostatiskie līdz 280 MPa (*ECE R134* vai *ISO/TR 15916:2015*).

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DŪMGĀZU KONDENSATORA EFEKTIVITĀTES PAAUGSTINĀŠANA, IZMANTOJOT SILTUMSŪKŅU TEHNOLOGIJU

INCREASING THE FLUE GAS CONDENSER EFFICIENCY USING
HEAT PUMP TECHNOLOGY

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Latvijas kurss uz klimatneitralitāti, kas jāpanāk 2050. gadā, ir saistīts arī ar siltumenerģijas brīvā tirgus atvēršanu [1]. Labs piemērs ir Rīgas pilsēta – AS "Rīgas Siltums" ražo vien 30 % nepieciešamās siltumenerģijas šādā proporcijā: 52 % atjaunojamie energoresursi (koksne), 48 % dabasgāze. Savukārt 70 % siltumenerģijas no kopējās bilances tiek iepirkta no neatkarīgajiem siltumenerģijas ražotājiem iknedējas iepirkumu procedūrās [2].

Brīvās konkurences apstākļos ļoti būtisks ir ne tikai katla lietderības koeficients, bet arī tas, lai sistēma būtu aprīkota ar dūmgāzu ekomaizeru-kondensatoru. Iekārta samazina dūmgāzu temperatūru līdz tīkla ūdens atgaitas temperatūrai, atgūstot līdz pat 30 % no kopējās katla jaudas, samazinot kopējo kurināmo patēriņu un CO₂ izmešus.

Dūmgāzu dziļāku dzesēšanu ierobežo siltumtīklu atgaitas ūdens temperatūra. Lai varētu pilnībā izmantot aizejošo dūmgāzu siltuma potenciālu, tiek izmantota siltumsūkņu tehnoloģija. Patlaban Latvijā ir realizēti projekti, kuros izmantoti lieljaudas absorbcijas tipa siltumsūkņi.

Siltumsūkņiem galvenais efektivitātes rādītājs ir *COP*.

Absorbcijas tipa siltumsūkņa efektivitāti aprēķina pēc šādas formulas [3]:

$$COP = \frac{Q_{kop}}{Q_{kop} - Q_{ieg}}. \quad (1.1.)$$

MECHANICAL ENGINEERING TECHNOLOGY AND HEAT ENGINEERING

Lai varētu realizēt siltuma noņemšanu no dūmgāzēm, dūmgāzu kondensators tiek veidots divās pakāpēs. Pirmajā pakāpē tiek pazemināta dūmgāzu temperatūra līdz siltumtīklu atgaitas ūdens temperatūrai (jāatzīmē, ka dūmgāzu temperatūra vidēji par 3–4 °C grādiem augstāka nekā siltumtīklu atgaitas ūdens temperatūra, kas ir saistīts ar siltumpāreju), arī otrajā kondensatora pakāpē dūmgāzes tiek dzesētas ar industriālo siltumsūknī.



1. att. Absorbcijas tipa siltumsūkņa rūpnīcas testi (V. Žentiņa fotogrāfija).

Tika veiks eksperimenta, biokurināmajam katlam ar nominālo jaudu 20 MW uzstādot divpakāpju dūmgāzu kondensatoru, turklāt otrajā kondensatora pakāpē dūmgāzes tika dzesētas ar absorbcijas tipa siltumsūkņi. Siltumsūkņa kopējā uzstādītā jauda ir 4720 kW. Katlam strādājot dažādās siltumslodzēs nevienmērīgos režīmos, *COP* rādītājs svārstījās 1,68–1,75, papildus no dūmgāzēm iegūstot līdz pat 2,49 MW siltumenerģijas. Eksperimenta dati parādīja, ka dūmgāzu temperatūra tika samazināta līdz 23,9 °C. Samazinoties dūmgāzu temperatūrai, dati uzrādīja kondensācijas kapacitātes paaugstināšanos. Atdzesējot dūmgāzes veidojas kondensāts, kura temperatūra bija 38,4 °C un apjoms svārstījās no 3 m³/h līdz 5 m³/h.

Secinājumi

1. Izmantojot reāla objekta analīzi ar absorbcijas tipa siltumsūkni, tika aprēķināts COP koeficients, kas svārstījās 1,68–1,73 robežās, iegūstot līdz 2,49 MW siltumenerģijas.
2. Veicot pētījumu par siltumavota efektivitātes paaugstināšanu, secināts, ka dūmgāzu dzīlā dzesēšana rada kondensāta veidošanos kondensatorā. Eksperimenta laikā tika konstatēts kondensāts, kas tiek novadīts kanalizācijā. Kondensāta temperatūra bija 38,4 °C un apjoms svārstījās no 3 m³/h līdz 5 m³/h.
3. Dažādu darba režīmu mainīgie parametri parādīja, ka nepieciešams turpināt papildu datu ievākšanu, lai varētu izveidot līknes, piemēram, nemot vērā siltumtīkla parametrus, kurināmā sastāvu un mitrumu.

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KAVITĀCIJAS IETEKMES PĒTĪJUMI UZ ZEMA SPIEDIENA ŠĶIDRO METĀLU AR BRĪVU VIRSMU

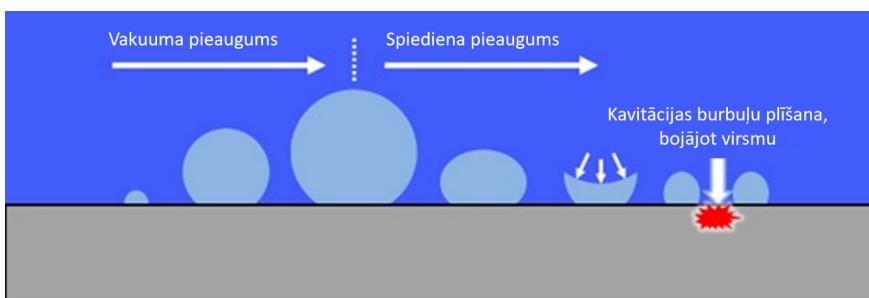
INVESTIGATION OF CAVITATION INFLUENCE ON LOW PRESSURE
LIQUID METAL WITH FREE SURFACE

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Mūsdienās zinātne nav iedomājama bez dažāda veida izotopiem. Izotopi nepieciešami medicīnā (vēža ārstēšanā), materiālu pētīšanā un citās zinātnes nozarēs. Lai iegūtu izotopus, ar protona staru tiek šauts pa dažādu materiālu mērķiem. Vieni no šādiem mērķiem ir šķidrie metāli. Lai efektīvāk iegūtu izotopus no šķidrā metāla mērķa, ir jāpalielina virsmas laukums, un tas tiek panākts, brīvi krītošu šķidrā metāla plūsmu veidojot daudzu sīku pilienu strūklās. Šīs sīku pilienu strūklas tiek veidotas vakuumā un noslēdzas tvertnē ar brīvo virsmu. Zem šīs tvertnes ir šķidrā metāla staba augstums, kas veido zemu virsspedienu sūknī ieejā, kā rezultātā sūknis nespēj sasniegt vajadzīgos tehniskos parametrus un veidojas kavitācija.



1. att. Kavitācijas burbuļu veidošanās un sabrukšana pie virsmas [1].

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Pētījuma mērķis ir apzināt iepriekš zināmos pētījumus par kavitācijas ietekmi uz zema spiediena šķidra metāla kontūru darbību ar brīvu virsmu.

Pētījumā apskatīti pieejamie zinātniskie avoti par kavitācijas pētījumiem šķidro metālu kontūros ar un bez brīvās virsmas.

Zinātniskajos rakstos redzams, ka kavitācija ir gāzveida burbuļu veidošanās šķidrumā zema spiediena apgabalos. Šiem burbuļiem pārvietojoties uz zonu, kur spiediens palielinās, tie plīstot izdala enerģiju, kas absorbējas šķidruma tilpumā vai rada bojājumus uz virsmām. Kavitācija izraisa sūkņu veikspējas samazināšanos, darbības mūža ilguma samazināšanos, troksni un mehāniskās vibrācijas.

Lielā daļā zinātnisko rakstu ir pētīti kavitācijas procesi ūdenī un dažādos šķidrumos. Taču maz sastopami ir raksti par kavitācijas detektēšanu un mērišanu šķidrā metāla kontūros un sūkņos.

Ir redzams, ka nepieciešams izpētīt šķidrā metāla sūkņu darbību pie zemām spiedienā vērtībām sūkņa ieplūdē, lai noteiktu kavitācijas ietekmi uz elektromagnētiskās indukcijas sūkņiem ar rotējošiem pastāvīgajiem magnētiem.

Atsauce

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PRIEKŠĒJĀ RATA AIZMUGURĒJĀS SVIRAS LŪZUMA METALOGRĀFISKĀ EKSPERTĪZE

METALLOGRAPHIC EXPERTISE OF FRONT WHEEL
REAR LEVER FRACTURE

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PAR AUTOMAŠĪNAS AUDI A6 CEĻA SATIKSMES NEGADĪJUMU 2021. GADĀ

Pēc Valsts policijas Vidzemes reģiona pārvaldes Valmieras iecirkņa kriminālpolicijas nodaļas pieprasījuma tika veikta metalogrāfiskā ekspertīze kriminālprocesa lēmuma pieņemšanai. Ekspertīzes priekšmets – transporta līdzekļa *AUDI A6* priekšējā labā riteņa aizmugurējā svira.

Celu satiksmes negadījuma aprakstā minēts, ka vadītājs izjutis sitienu pa priekšējo riteni, mašīna samesta un iebraukusi ceļa pretējās putas grāvī. Turpretim veiktajā transporta līdzekļa apskates laikā konstatēts, ka cietusi priekšējā labā riteņa riepa. Pēc rata noņemšanas konstatēts, ka pārlūzusi priekšējā labā rata apakšējā aizmugurējā svira. Lūzuma vieta ir svaiga, bez korozijas, metāla krāsa viscaur vienādā tonī.

1. attēlā redzams priekšējā rata aizmugures sviras kopskats pēc lūzuma (lūzuma vieta, triecienu virziens). Ap lūzuma vietu deformācijas rezultātā atdalījusies virsmas oksīdu kārtiņa (gaišākā krāsā).



1. att. Priekšējā labā rata aizmugurējā svira (kopskats).

Lai atbildētu uz jautājumu, vai transporta līdzekļa AUDI A6 priekšējā rata aizmugurējā svira sabojāta pirms ceļu satiksmes negadījuma, vai tā laikā, veikta lūzuma tehniskā analīze.

Lūzuma vietas apskate uzrāda, ka lūzums veidojies vairākās pakāpēs. 2. attēlā redzamās priekšējā rata aizmugurējās sviras lūzums vispirms iekšējā malā veido līdz 2 mm dziļu iepriekšējo iegriezumu A₀ un noguruma lūzumu (A). Šeit virsmas savstarpēji noslīpējušās spīdīgas.



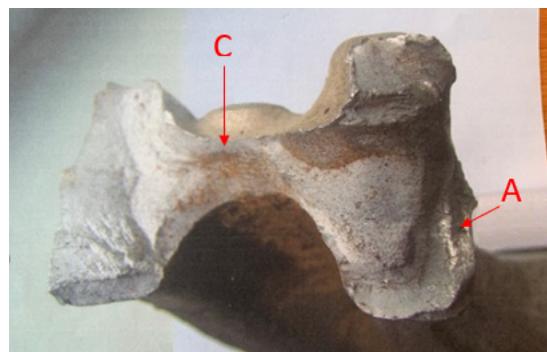
2. att. Priekšējā rata aizmugurējās sviras lūzuma attēls.

MECHANICAL ENGINEERING TECHNOLOGY AND HEAT ENGINEERING

Atsevišķā daļā (B) redzamas korozijas pēdas. Radītais šķērsgriezuma samazinājums nerada būtisku sviras stiprības samazinājumu un var turpināt kalpot ekspluatācijas gaitā.

Ja automašīnas priekšējo riteņu stūres konstrukcija ekspluatācijas gaitā saņem pēkšņu triecienu, tad noguruma plaisa var kalpot par sprieguma koncentratoru un attīstās līdz pilnīgai sagraušanai. Stiepes spriegumi veido trauslu lūzumu (C). Tā virsma ir gluda, graudaina, bez plastiskās deformācijas.

3. attēlā redzama sviras lūzuma virsma no pretējās puses, salīdzinot ar 2. attēlu. Arī šeit var redzēt noguruma lūzumu (A) klātbūtni, kā arī korozijas pēdas.



3. att. Sviras lūzuma virsmas attēls no pretējās puses, salīdzinot ar 2. attēlu.

Pilnīga priekšējā riteņa aizmugurējās sviras sagrūšanas ir notikusi trieciena laikā, stūres mehānisms ir zaudējis darbības spējas un notikusi automašīnas avārija.

Otrs jautājums ir – kāda ir konkrētās detaļas materiāla kvalitāte, vai iepriekš ir bijušas mehāniskas iejaukšanās?

Aizmugurējā svira ar Nr. 470.407.694C ir oriģināla un nav bijusi pakļauta mehāniskam remontam. Lūzuma virsmā nav ieslēgumu, tā ir smalkgraudaina. Tas liecina par tās kvalitāti.

Virsmā papildus novērots neliels iedobums, kas iepriekš varēja radīt darbs ar lokanu slīppripu vai arī tas varēja rasties saskarsmē ar cietu ķermenī brauciena laikā. No šī iedobuma A0 vēlāk ir attīstījies sākotnējais noguruma lūzums A.

Atsauce

1. J. A. Geller, G. I. Pogodin-Aleksee, A. G. Rahšadt "Metālzinātne", "Metallurgija", Maskava, 1967 g., 131–135 lpp.